

# Determination of the mechanical properties of polymeric microneedles by micromanipulation

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1 **Determination of the mechanical properties of polymeric microneedles by micromanipulation**

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8 **Abstract:**

9 Precise characterization of the mechanical properties of polymeric microneedles is crucial for their  
10 successful penetration into skin and delivery of the loaded active ingredients. However, most available  
11 strategies for this purpose are based on compression of the whole patch, which only provide the  
12 average rupture force of the needles and can not give information on the variations across individual  
13 microneedles in the patch. In this study, we determined the mechanical strength of individual  
14 microneedles of two types of hyaluronic acid microneedles with or without loaded model drugs using a  
15 micromanipulation technique. The applied force as a function of displacement of the microneedles was  
16 recorded, which was used to determine the rupture displacement, rupture force, and then to derive and  
17 calculate normal stress-deformation curve, rupture stress and Young's modulus of individual  
18 microneedles. The obtained data suggest that the molecular weight of the polymer and the loading of  
19 drug into the microneedles can significantly affect the rupture behavior and mechanical properties of  
20 the microneedles, which provides a foundation for preparing sufficiently strong microneedles for  
21 controlled drug delivery.

22 **Keywords:** Mechanical properties, Polymeric microneedles, Micromanipulation, Rupture force, Normal  
23 stress

24 **1. Introduction**

25 Polymeric microneedles have been widely investigated for drug delivery, medical diagnosis and health  
26 monitoring [1-3]. They can pierce skin barrier in a non-invasive and pain-free way as they do not touch  
27 nerves or blood capillaries inside the skin during application. As compared to solid and hollow  
28 microneedles made of glass or metal, polymeric microneedles made of dissolving or biodegradable  
29 polymers also hold advantages including resulting in no hazardous waste after administration and  
30 easiness for modulation of release properties of the loaded ingredients [4,5]. However, polymer based  
31 microneedles have a relatively weak mechanical strength, which may cause the breakage or bending of  
32 the microneedles during the insertion of skin, resulting in an insufficient penetration [6,7]. Direct and  
33 precise measurement of the mechanical properties of polymeric microneedles is necessary for ensuring  
34 their successful application especially in case of industrial mass production. Besides, emerging new types  
35 of advanced microneedles have been designed and investigated in the past decade, such as bio-  
36 responsive microneedles [8,9], core-shell structured microneedles [10,11], and hydrogel microneedles

37 [12]. Precise characterization of the mechanical properties of these microneedles is crucial for their  
38 possible translation as the complex composition and design could significantly affect their mechanical  
39 strength.

40 In many microneedle studies, the mechanical strength of microneedle patch was investigated by  
41 compressing the whole patch against a flat surface, after which the rupture force of single microneedles  
42 was calculated by dividing the total rupture force by the number of needles [7,13]. This strategy is not  
43 adequate since it can not identify the possible variations of the mechanical properties among the  
44 microneedles across the patch. The mechanical properties obtained are also limited to the rupture force  
45 of the bulk patch. In other microneedle studies, their mechanical properties were not directly measured  
46 but were instead reflected by their skin penetration efficiency. The small holes in the skin generated by  
47 microneedle penetration were normally stained and visualized for calculating the skin penetration  
48 efficiency [14,15]. However, this method gives no quantitative results for the mechanical property of  
49 microneedles. Atomic force microscopy (AFM) has also been used to measure the mechanical properties  
50 of microneedles. However, their indenting depth or force is limited to nanoscale measurement [16,17].

51 Micromanipulation is an experimental technique that was first developed for measurement of the  
52 bursting force of single mammalian cells [18]. This technique was then extended to analyze other  
53 biological or non-biological micro-particles, including microcapsules and microspheres [19-21]. Micro-  
54 particles can be rested on a glass slide or in a chamber and then compressed by using a cylindrical probe  
55 with a diameter larger than the micro-particles. As compared to other bulk methods for mechanical  
56 property characterization, the technique can record the force-displacement curves generated from  
57 compression of single micro-particles, which can be further used to extract important mechanical  
58 property parameters including rupture force, displacement at rupture and normal rupture stress [22,23].  
59 The generated data can also be used to determine the intrinsic material properties of the samples, such  
60 as Young's modulus, yield stress and stress and strain at rupture by analytical or numerical modeling  
61 [19].

62 Herein, we investigated the mechanical properties of two types of home-made hyaluronic acid (HA)  
63 based dissolving microneedles with or without loaded model drugs of lidocaine hydrochloride and  
64 bupivacaine hydrochloride with a micromanipulation technique. Both of them are hydrophilic small-  
65 molecule drugs and used as local anesthetic of the amino amide type in clinic. The use of  
66 micromanipulation allows the precise measurement of the mechanical properties of individual  
67 microneedles. The force-displacement data were first recorded with this technique, which were used to  
68 determine the rupture displacement and rupture force, and then to derive normal rupture stress and  
69 Young's modulus of the microneedles. The influences of molecular weight of HA and the drug loading on  
70 the mechanical properties of the microneedles were investigated.

## 71 **2. Method and materials**

### 72 *Materials*

73 HA with a molecular weight of 10-kDa and 300-kDa were purchased from Bloomage Biotech (Jinan,  
74 China). Lidocaine hydrochloride and bupivacaine hydrochloride as model drugs were received as gifts

75 from West China Hospital of Sichuan University (Chengdu, China). Rhodamine B dye was purchased from  
76 Meilunbio (Dalian, China). Milli-Q water was used for the preparation of all solutions. All the other  
77 reagents used were of analytical grade.

#### 78 *Fabrication of HA based microneedles*

79 Different types of fabricated quadrangular pyramid microneedles (700  $\mu\text{m}$  height, 300  $\mu\text{m}$  base width  
80 and 600  $\mu\text{m}$  center-center spacing) in a 10 by 10 array on a back plate of  $0.9 \times 0.9 \text{ cm}^2$  are summarized  
81 in Table 1. All of the microneedles were fabricated by using a micro-molding method as previously  
82 reported [24]. Briefly, polydimethylsiloxane (PDMS) molds were first duplicated from stainless steel  
83 microneedle molds. Next, 30  $\mu\text{l}$  HA solution dissolved with or without drugs was loaded into the cavities  
84 of a PDMS mold by using pressurized air to form the needles of the patch. After drying for 0.5 h in  
85 anhydrous silica gel environment, 40  $\mu\text{l}$  of blank HA solution was added into the mold to form the back  
86 plate of the patch. Finally, the microneedle patches were dried for another 4 h and subsequently peeled  
87 off from the mold. The fabricated patches were stored in dry environment for further characterization.  
88 To fabricate microneedle patches for visualization by confocal microscopy, rhodamine B dye was  
89 dissolved in HA solution for the fabrication.

90 **Table 1.** Different types of microneedle (MN) patches characterized in the current study

MN type	Matrix	Abbreviation name
1	10-kDa HA	Blank HA (10-kDa) MN
2	300-kDa HA	Blank HA (300-kDa) MN
3	Bupivacaine + 10-kDa HA (Weight ratio 1:2)	Bupi HA (10-kDa) MN
4	Lidocaine + 300-kDa HA (Weight ratio 4:5)	Lido HA (300-kDa) MN

91

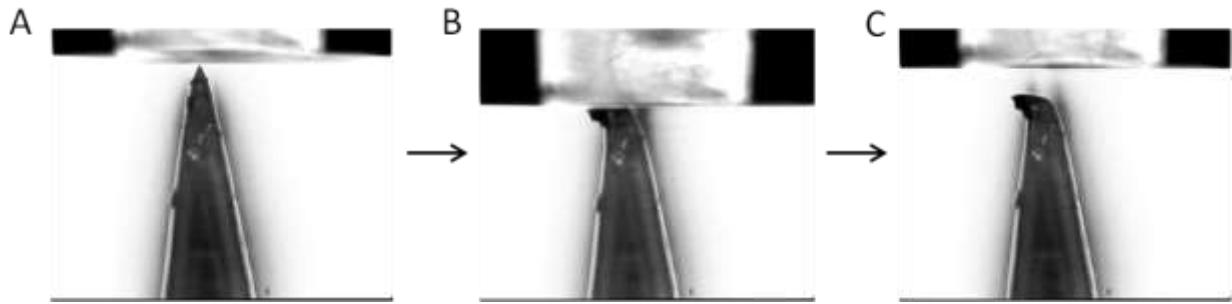
#### 92 *Visualization of microneedles*

93 The surface morphology of the fabricated microneedles was visualized using scanning electron  
94 microscopy (SEM) (JEOL JSM-7500F, Japan) with an operation voltage of 15 kV. The microneedle patches  
95 were fixed on SEM stub and coated with a thin layer of carbon for the visualization. The microneedles  
96 containing rhodamine B were visualized by confocal laser scanning microscopy (ZEISS CLSM, Overkochen,  
97 Germany) by scanning them with a depth resolution of 5  $\mu\text{m}$ /step using a 4 $\times$  Plan Apo objective with a  
98 4 $\times$  magnification.

#### 99 *Micromanipulation*

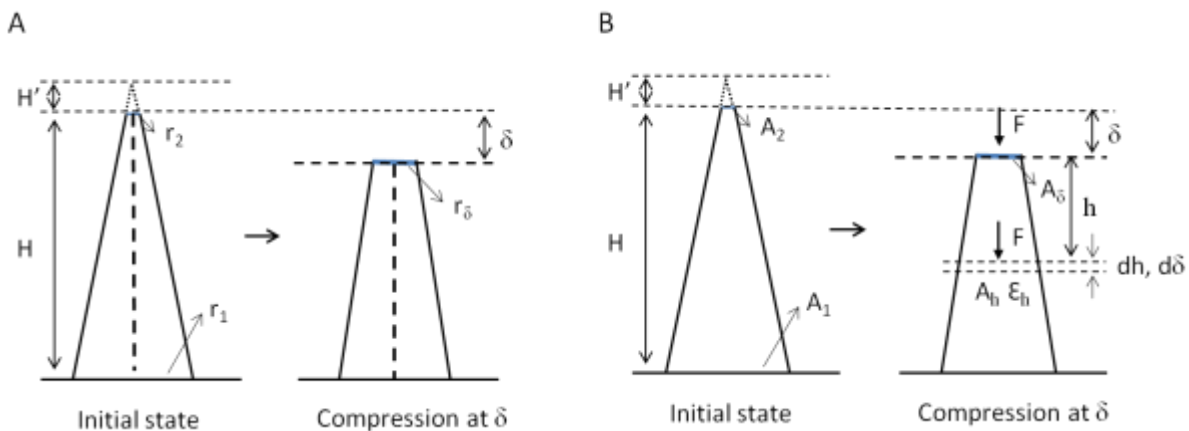
100 The detailed working principles of the micromanipulation technique have been introduced in previous  
101 studies [18,21]. Briefly, to start the compression, the microneedle patches were placed on the stage of  
102 the micromanipulation instrument equipped with a force transducer (Model GSO-10, Transducer  
103 Techniques, LLC, USA). An optical glass rod made of Borosilicate with a flat end of a diameter of 100  $\mu\text{m}$

104 was mounted onto the force transducer for the compression. Single microneedles were compressed  
 105 between the stage and the glass probe with a compression speed of  $2 \mu\text{m/s}$ . The applied force on single  
 106 microneedles as a function of their displacement was recorded. One typical example of different status  
 107 of the single microneedles during compression is shown in Figure 1. The compression process is shown  
 108 in supplementary video. For each type of microneedles, 30 needles were measured.



109  
 110 **Figure 1.** One example of different status of individual microneedle during the compression measurement. A:  
 111 Before compression, B: Fracture of microneedles, C: After compression.

112 *Calculation of the normal stress from compression force*



113  
 114 Figure 2. Schematic diagram for modeling of stress (A) and Young's modulus (B). Abbreviations: H: initial  
 115 height of a microneedle. H': height of the missing tip when assuming the microneedle is perfectly sharp.  
 116  $r_1$ : half side length of a quadrangular microneedle base.  $r_2$ : initial half side length of the quadrangular  
 117 microneedle tip.  $\delta$ : displacement of the microneedle.  $r_\delta$ : half side length of contact surface at a  
 118 compressing displacement of  $\delta$ .  $A_1$ : surface area of the quadrangular microneedle base.  $A_2$ : surface area  
 119 of the quadrangular microneedle tip.  $A_\delta$ : contact surface area at a compressing displacement of  $\delta$ . F:  
 120 compression force at a displacement of  $\delta$ . h: height of the microneedle (from top of the microneedle)  
 121 until sectional area of  $A_h$ .  $\epsilon_h$ : deformation at a height of h. dh: the length of the element at h.  $d\delta$ :  
 122 displacement at h.

123

124 The normal stress was calculated from compression force using modeling as shown in Figure 2A. In this  
125 modeling, we ignored the volume change of microneedle and the deformation of microneedle base  
126 during compression (Figure 2A).

127 The volume of a single microneedle can be calculated by using dimension parameters as below (Eq.1).

$$128 \quad V=4/3r_1^2(H+H')-4/3r_2^2H' \quad (\text{Eq.1})$$

129 where V is the volume of the microneedle,  $r_1$  is half side length of the quadrangular microneedle base, H  
130 is initial height of the microneedle,  $H'$  is height of the missing tip when assuming the microneedle is  
131 perfectly sharp and  $r_2$  is half initial side length of the quadrangular microneedle tip.

132 According to geometric similarity (Eq.2)

$$133 \quad H'/r_2=(H+H')/r_1 \quad (\text{Eq.2})$$

134 By combining microneedle volume (Eq.1) and geometric similarity (Eq.2) equations, we can eliminate  $H'$   
135 from the equations, and  $r_2$  can be calculated from parameters which are already known (Eq.3).

$$136 \quad r_2 = 1/2(-r_1 + \sqrt{3V/H - 3r_1^2}) \quad (\text{Eq.3})$$

137 After replacing  $r_2$  with  $r_\delta$  and replacing H by  $H - \delta$  at a compression displacement of  $\delta$ ,  $r_\delta$  can be obtained  
138 (Eq.4).

$$139 \quad r_\delta = 1/2(-r_1 + \sqrt{3V/(H - \delta) - 3r_1^2}) \quad (\text{Eq.4})$$

140 Finally, the normal stress can be calculated by dividing compression force by the contacting area  $4 r_\delta^2$   
141 (Eq. 5).

$$142 \quad \text{Normal stress} = F/(4r_\delta^2) \quad (\text{Eq.5})$$

143

#### 144 *Young's modulus calculation*

145 Young's modulus was calculated from compression force using a modeling as shown in Figure 2B.

146 We assumed that the microneedle was linearly elastic under a small deformation and only considered  
147 the longitudinal deformation. At a height of h (from the top of the microneedle) with the sectional area  
148  $A_h$ , the deformation  $\epsilon_h$  can be calculated by the stress-strain equation (Eq.6).

$$\epsilon_h = \frac{F}{EA_h} \quad (\text{Eq. 6})$$

149 Where F is the compression force, E is Young's modulus.

150  $\epsilon_h$  can also be given by

$$\epsilon_h = \frac{d\delta}{dh} \quad (\text{Eq.7})$$

151 where  $d\delta$  is the displacement at h and  $dh$  is the length of the element at h.

152 From geometric similarity

$$\frac{A_h}{A_1} = \left( \frac{h + H'}{H + H'} \right)^2 \quad (\text{Eq. 8})$$

153 Thus,

$$A_h = \left( \frac{h + H'}{H + H'} \right)^2 A_1 \quad (\text{Eq.9})$$

154 The overall displacement  $\delta$  can be calculated by integrating  $d\delta$  over the microneedle as

$$\delta = \int_0^H d\delta \quad (\text{Eq.10})$$

155 Submitting Eq.6 to 9 into Eq.10 leads to

$$\delta = \frac{F}{A_1 E} \int_0^H \frac{1}{\left( \frac{h + H'}{H + H'} \right)^2} dh \quad (\text{Eq.11})$$

156 which can be written as follows

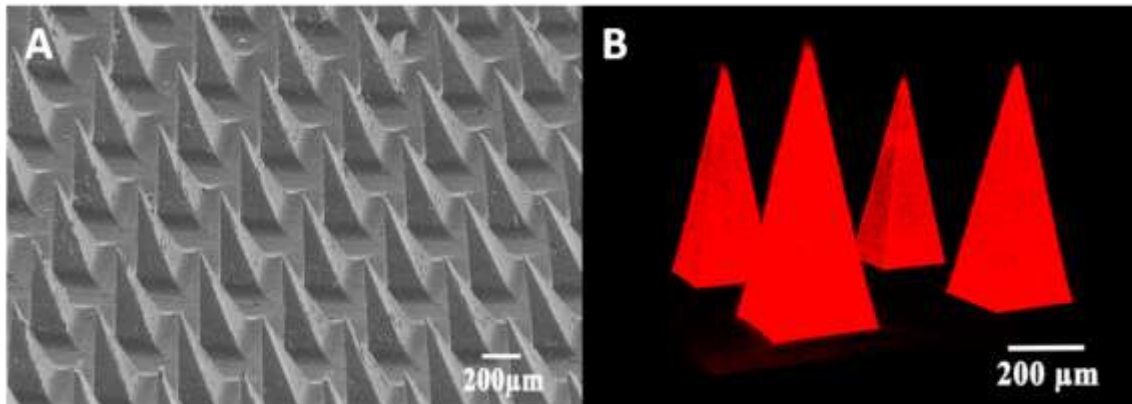
$$E = \frac{F(H+H')H}{\delta A_1 H'} \quad (\text{Eq.12})$$

157 *Statistical analysis*

158 In total 30 microneedles from each type of samples were analyzed. All the data of the mechanical  
159 properties were analyzed by one way ANOVA with Turkey's post-test using GraphPad Prism software  
160 (version 5.02). The significance levels were set at \* $p < 0.05$ , \*\* $p < 0.01$ , and \*\*\* $p < 0.001$ .

### 161 3. Results

#### 162 Morphology of microneedles

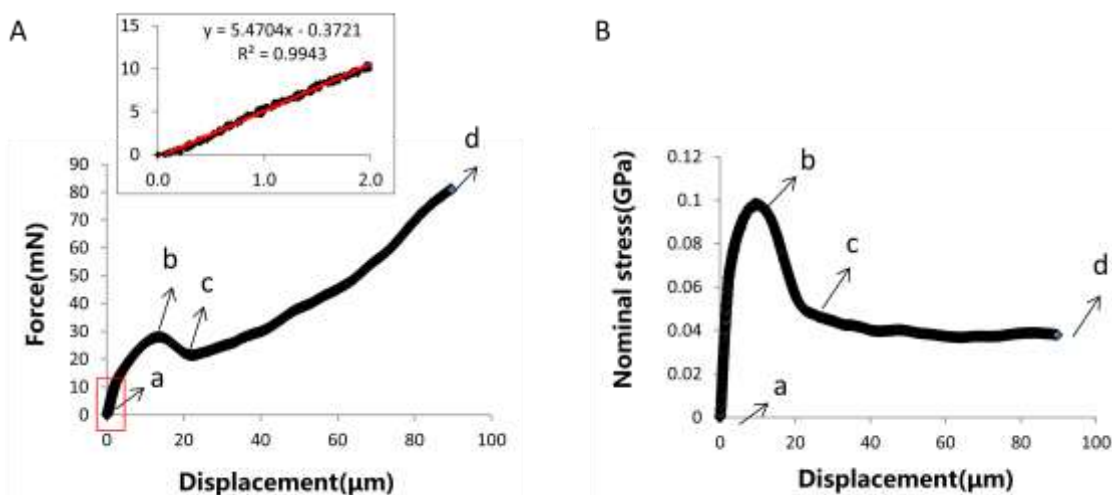


163

164 **Figure 3.** Representative microscopy images of the fabricated microneedles. A: Scanning electronic microscopy  
165 (SEM) images. B: Confocal laser scanning microscopy (CLSM) images.

166 The microscopy images of representative microneedles in a patch are shown in Figure 3. SEM (Figure 3A)  
167 and CLSM (Figure 3B) images show that the obtained microneedles had a regular quadrangular pyramid  
168 shape and sharp tips.

### 169 **Force versus displacement and normal stress versus displacement curves**

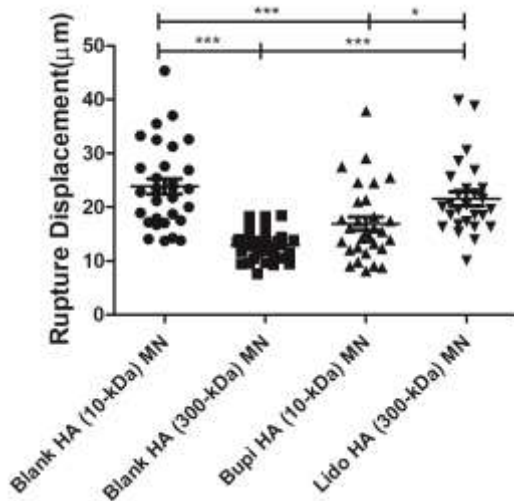


170  
171 **Figure 4.** Typical force versus displacement (A) and the corresponding normal stress versus displacement (B) curves  
172 obtained from compression of a single microneedle.

173 A typical relationship of the force applied on an individual microneedle and the corresponding normal  
174 stress as a function of the probe moving distance (displacement) is shown in Figure 4A and 4B,  
175 respectively. Immediately after point-a the probe touched the microneedle tip, the force increased until  
176 point-b at which the microneedle ruptured (Figure 4A). As a result, the force decreased to point-c and  
177 increased again until reaching the detection limitation of the sensor at point-d. Figure 4B shows the  
178 derived normal stress-displacement curve. In contrast, the normal stress increased from point-a to peak  
179 point-b when the needle ruptured, after which the stress decreased to point-c and reached a plateau  
180 where it stayed roughly the same until the end of the measurement (point-d). From these curves, the  
181 rupture force and rupture stress (normal stress at rupture) can be determined (point-b). The rupture  
182 displacement is identified as the distance that the probe travels from point-a until point-b when the  
183 microneedle ruptures. The video of the compression process is shown in Supplementary information.

### 184 **Summary of the rupture displacements**

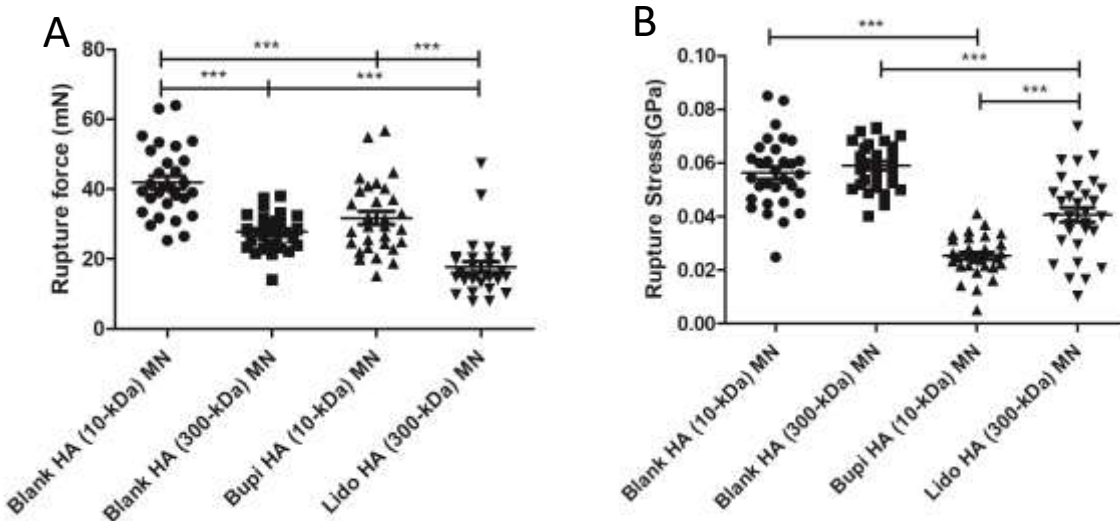




185  
186 **Figure 5.** Rupture displacements of various types of microneedles. \*:p<0.05; \*\*:p<0.01;\*\*\*:p<0.001.

187 We summarized the rupture displacements of different types of microneedles in Figure 5. It is shown  
188 that Blank HA (300-kDa) MN patch had the narrowest distribution of rupture displacement among all  
189 patches. Specifically, the characterized patches showed an average rupture displacement between  $12.9$   
190  $\pm 2.8 \mu\text{m}$  (Blank HA (300-kDa) MN) and  $23.8 \pm 7.9 \mu\text{m}$  (Blank HA (10-kDa) MN). The rupture displacement  
191 of microneedles made of 10-kDa HA ( $23.8 \pm 7.9 \mu\text{m}$ ) was significantly higher than that of microneedles  
192 made of 300-kDa HA ( $12.9 \pm 2.8 \mu\text{m}$ ). The drug loaded patch showed either a lower (Bupi HA (10-kDa)  
193 MN,  $16.9 \pm 6.9 \mu\text{m}$ ) or higher (Lido HA (300-kDa) MN,  $21.6 \pm 6.8 \mu\text{m}$ ) rupture displacement than the  
194 corresponding blank HA patches.

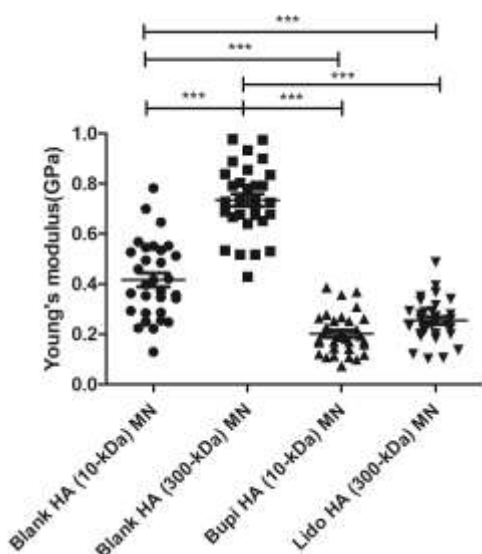
195 **Summary of rupture forces and stresses**



196  
197 **Figure 6.** Rupture forces and stresses of different types of microneedles. \*:p<0.05; \*\*:p<0.01;\*\*\*:p<0.001.

208 The rupture forces and stresses of different types of microneedle patches are summarized and  
 209 presented in Figure 6. In case of rupture force, blank microneedles made of 10-kDa HA showed a mean  
 210 rupture force of  $42.0 \pm 9.9$  mN, which was significantly higher than that of the microneedles made of  
 211 300-kDa HA ( $27.7 \pm 5.2$  mN). The loading of drug in HA (10-kDa) MN and HA (300-kDa) MN significantly  
 212 decreased the rupture force of the corresponding microneedles to  $31.7 \pm 10.3$  mN and  $17.7 \pm 8.3$  mN,  
 213 respectively. As for rupture stress, by contrast, blank microneedles made of 10-kDa and 300-kDa HA  
 214 showed similar rupture stresses of  $0.072 \pm 0.017$  GPa and  $0.075 \pm 0.011$  GPa. The loading of drug into  
 215 the microneedles made of 10-kDa and 300-kDa HA significantly decreased the rupture stresses by  
 216 around 50% and 25%, respectively.

217 **Young's modulus of the microneedles**



218 **Figure 7.** Calculated Young's modulus of different types of microneedles. \*:p<0.05; \*\*:p<0.01;\*\*\*:p<0.001

219 The relationship between force and displacement was found be linear for displacements up to  $2 \mu\text{m}$ , as  
 220 predicted by the elastic model (Eq.9), which was used to calculate the Young's modulus of different  
 221 microneedles via liner regression (see insert of Figure 4A) and the Young's modulus values are presented  
 222 in Figure 7. Interestingly, blank microneedles made of 300-kDa HA showed a significantly higher Young's  
 223 modulus than that made of 10-kDa HA. The incorporation of drugs into microneedles also significantly  
 224 decreased the Young's modulus of the microneedles: the loading of bupivacaine into 10-kDa HA MN  
 225 slightly decreased the Young's modulus from  $0.42 \pm 0.15$  to  $0.20 \pm 0.08$  GPa while the loading of  
 226 lidocaine into 300-kDa HA microneedles robustly decreased the Young's modulus from  $0.73 \pm 0.14$  to  
 227  $0.25 \pm 0.09$  GPa.

228 **4. Discussion and conclusions**

229 One aim of this study was to investigate the potential of the micromanipulation technique for  
 230 characterization of the mechanical properties of polymeric microneedles. We therefore fabricated HA  
 231 microneedles with two different molecular weights with or without loaded model drugs. We measured

223 and compared their mechanical properties with micromanipulation. Our results showed that the micro-  
224 manipulation technique is a powerful tool for precise and comprehensive characterization of the  
225 mechanical properties of polymeric microneedles. The change of the compression force as a function of  
226 displacement behavior of single microneedles can be precisely recorded during the compression. With  
227 this information as well as mathematic modeling, the normal stress at the contact area between the  
228 force probe and the microneedle as a function of its displacement and several important  
229 mechanical/material property parameters including displacement at rupture, rupture force and Young's  
230 modulus can be analyzed and calculated. As compared to bulk compression strategies,  
231 micromanipulation has at least two advantages. Firstly, in comparison with different bulk compression  
232 strategies as previously reported, which assume all microneedles within one patch have the same  
233 mechanical strength, micro-manipulation can directly and precisely measure the rupture behavior of  
234 individual microneedles, therefore can give information on the uniformity of the microneedle strength  
235 across the patch. Our data indeed showed that the mechanical property parameters of microneedles  
236 across one patch can vary significantly (Figure 5-7). Secondly, the bulk compression strategies only  
237 measure rupture forces to evaluate the mechanical property of microneedle patches. Additionally, we  
238 compressed the whole patch of Blank HA (10-kDa) MN with a traditional strategy. We even could not  
239 observe a clear rupture behavior of the patch (Supplementary Figure 1). In micro-manipulation, as  
240 shown in our results, other parameters including displacement at rupture, rupture stress as well as  
241 intrinsic material properties such as Young's modulus can also be measured. Therefore, the  
242 micromanipulation technique can be used to comprehensively evaluate the mechanical properties of  
243 microneedles within one patch and to compare the difference in mechanical properties among different  
244 type of patches.

245 During the measurement, we obtained in total four mechanical/material property parameters that can  
246 reflect the mechanical strength of microneedles, including displacement at rupture, rupture force,  
247 rupture stress and Young's modulus. Young's modulus can reflect the stiffness of the measured sample,  
248 while displacement at rupture describes the maximum deformation level before the rupture of  
249 microneedles [25,26]. Both of these two indicators can be used for comparison of the mechanical  
250 strength of different microneedles when the measured microneedles are made of the same materials.  
251 This is indicated by our results that although Blank HA (10-kDa) MN showed a lower Young's modulus  
252 (Figure 7), they showed a higher rupture force as compared to the Blank HA (300-kDa) MN (Figure 6). On  
253 the other hand, Blank HA (10-kDa) MN showed a much higher rupture displacement than Blank HA (300-  
254 kDa) MN (Figure 5), but they own a similar rupture stress to the later one (Figure 6).

255 Most of the reported studies made use of rupture force of the microneedles for evaluation of the  
256 mechanical strength of microneedles [8,10,27,28]. However, our results showed that rupture force and  
257 rupture stress results could give different trends. For example, based on rupture force results, Blank HA  
258 (10-kDa) MN is mechanically stronger than Blank HA (300-kDa) MN (Figure 6A). However, their  
259 mechanical strengths are similar according to rupture stress results (Figure 6B). Actually, the rupture  
260 force of microneedles could be significantly affected by the rupture displacement/contacting area at  
261 rupture. As a result, simple comparison of rupture force is not enough for the evaluation of the  
262 mechanical strength of different microneedle patches since it is not an intrinsic material property

263 parameter. In contrast, the rupture stress of microneedles can be considered to be an intrinsic material  
264 parameter, and our results indicated that it is a better indicator for the mechanical strength of polymer  
265 microneedles as compared to rupture force. The rupture stress results in the current study indicated  
266 that among the different type of microneedles that were investigated, Blank HA (10-kDa) MN and Blank  
267 HA (300-kDa) MN have a similar mechanical strength, but are all stronger than their corresponding drug  
268 loaded microneedle patches.

269 Our results in this study showed that both the molecular weight of the polymer and the loading of drug  
270 could significantly influence the mechanical properties of microneedles. We showed that blank  
271 microneedles made of 300-kDa HA had a significantly higher Young's modulus than that made of 10-kDa  
272 HA. This observed trend is consistent with a previous study showing that 50 kDa HA gel had a  
273 significantly higher elastic modulus than 10-kDa HA gel [29]. However, as discussed above, the greater  
274 Young's modulus of Blank HA (300-kDa) MN did not simply lead to a greater rupture stress because it  
275 was ruptured at smaller displacement/deformation. Sun et al. also [28] showed that the tensile strength  
276 of a material was proportional to Young's modulus for a given material, but the relationship can vary  
277 significantly with materials. On the other hand, our results showed that the loading of both lidocaine  
278 hydrochloride and bupivacaine hydrochloride significantly decreased the mechanical strength of the  
279 microneedles, as shown by the results of rupture force, rupture stress and Young's modulus of the  
280 microneedles (Figure 6-7). These results are consistent with previous reports. For instance, Park et al.  
281 [13] showed that the loading of calcein or bovine serum albumin into poly-lactide-co-glycolic (PLGA)  
282 based microneedles could significantly decrease the mechanical strength of the microneedles. These  
283 findings indicated that the mechanical strength of the microneedles was mainly contributed by the  
284 polymer matrix and the poor adhesion between the drug and polymer could cause mechanical failure  
285 sites for the microneedles [13]. Nevertheless, previous research has indicated that the ultimate rupture  
286 stress of skin was around  $0.027 \pm 0.009$  GPa. Our measured rupture stress results of microneedle  
287 patches except Bupi HA (10-kDa) are higher than this value, indicating that these microneedles may be  
288 strong enough for piercing of the skin [30,31]. It should be noted that the environmental factors  
289 including temperature and air humidity could significantly affect the mechanical property of HA  
290 microneedles. For instance, Wang et al. has observed that after storing HA microneedles in 60% relative  
291 humidity for 30 min at 25 °C, the mechanical strength of the microneedles became significantly lower  
292 [32]. The environmental factors will be closely monitored in our future measurements.

293 In conclusion, our studies showed that the micromanipulation technique is an effective tool for precise  
294 characterization of the mechanical properties of polymeric microneedles. The generated information  
295 could provide important information for rational design of polymeric microneedles with sufficient  
296 mechanical strength for skin penetration. The results also revealed that we should carefully evaluate the  
297 variations among the microneedles across the patches and pay attention to the influence of drug  
298 loading and molecular weight of the polymer microneedles on their mechanical strength.

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#### 304 **Conflict of interest**

305 There is no conflict of interest.

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